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# Micropatterning and Functionalization of Single Layer Graphene: Tuning Its Electron Transport Properties

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## <sup>3</sup> **Micropatterning and Functionalization of Single Layer Graphene:** <sup>4</sup> **Tuning Its Electron Transport Properties**

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#### 16 **Keywords:** Graphene patterning, Electron transport, Electrochemical bromination, Photolithography, All graphene device

17 **Abstract:** As a promising 2D material, graphene exhibits excellent 18 physical properties including single-atom-scale thickness and 19 remarkably high charge carrier mobility. However, its semi-metallic  $20$  nature with a zero bandgap poses challenges for its application in 21 high-performance field-effect transistors (FETs). In order to 22 overcome these limitations, various approaches have been explored 23 to modulate graphene's bandgap, including nanoscale confinement, 24 external field induction, doping, and chemical micropatterning. 25 Nevertheless, the stability and controllability still need to be  $26$  improved. In this study, we propose a feasible method that combines 27 electrochemical bromination and photolithography to precisely tune 28 the electron transport properties of single layer graphene (SLG). 29 Through this method, we successfully fabricated various brominated 30 SLG (SLGBr) micropatterns with high accuracy. Futher investigation 31 revealed that the electron transport properties of SLG can be32 conveniently tuned by controlling the degree of bromination. The 33 SLGBr exhibited a resistance, and have a decreasing conductance 34 with the bromination degree increasing. When the bromination 35 degree increased to a critical value, the SLGBr demonstrated 36 semiconducting characteristics. This research offers a prospective 37 route for the fabrication of graphene-based devices, providing 38 potential applications in the realm of microelectronics.

39 Graphene has been considered as the most 40 prospective material because of its excellent properties, 41 including single-atom-scale thickness<sup>[1]</sup>, extremely high <sup>interaction betwee</sup> 42 charge carrier mobility<sup>[2]</sup>, superior optical transparency<sup>[3]</sup>, e.g., van der 43 high thermal conductivity<sup>[4]</sup>, etc.. Recently, the mass chemical bond<sup>[</sup> 44 production of high-quality single layer graphene (SLG) at 45 wafer scale makes it qualified in various possible 46 applications such as micro/nano-electronics, sensors and 47 actuators, microelectromechanical system (MEMS), etc.<sup>[5-7]</sup>. has tuneabl 48 To this, the localized micropatterning and functionalization 49 of SLG at wafer scale is crucial in order to tune the physical on the in  $50$  and chamical proportion of a band gap electron. 50 and chemical properties, e.g., band gap, electron 51 conductivity, transmittance, wettability, etc., and to fulfil the 52 requirements of various industrial domains $[8]$ .

Great efforts have been made to fabricate SLG patterns. Physical methods can be classified into top-down and lic bottom-up categories<sup>[9]</sup>. The top-down methods are to remove the carbon atoms from pristine SLG wafer by the energy-beam techniques including the focused laser beam, ed electron beam, plasma beam, reactive ion beam, etc.<sup>[10]</sup>. The top-down methods can be realized by either direct writing or template forming combined with various lithography techniques. The bottom-up methods are to synthesize specially sized SLG on the patterned catalysts he such as copper, nickel, platinum, gold, etc.<sup>[11-13]</sup>. The most advantage of physical methods lies that the crystalline structure of SLG are not destroyed. Consequently, the intrinsic properties of SLG are kept very well. Nevertheless, the physical properties of SLG, e.g., the band gap, can be he regulated by the external physical field<sup>[14]</sup>. However, when the external physical field were dismissed, the band gap of SLG would go back to zero.

Comparing physic methods, chemical micropatterning can change the crystalline structure and thus, change the physical and chemical properties of SLG<sup>[8]</sup>. Chemical adsorption<sup>[15]</sup> or doping<sup>[16]</sup> have been adopted to tune the  $SLG$ 's properties by introducing the  $sp<sup>3</sup>-C$  defects. However, after thermal annealing, many of the functionalized SLG materials change back to be pristine due to the weak interaction between SLG and the adsorbates or dopants,  $\mathbf{e}$ ,  $\mathbf{e}$ ,  $\mathbf{e}$ ,  $\mathbf{e}$  and der Waals' force<sup>[17]</sup>, adsorption bond<sup>[18]</sup>, or even <sub>is</sub> chemical bond<sup>[19]</sup>. To this, we developed an in-situ electrochemically induced radical addition reaction to le functionalize SLG with high efficiency and stability<sup>[20, 21]</sup>. Here we report that the functionalized SLG micropatterns has tuneable electron transport properties, making SLG act as single-atomic-thickness resistor or rectifier, depending <sub>al</sub> on the introduced sp<sup>3</sup>-C defects density.

electrochemical micropatterning and functionalization procedures of SLG are depicted as in Scheme 1. First, a thin-layer viscosifier (HMDS) was spin-





**Scheme 1** Preparing processes of brominated SLG with patterns. (i) blue-shifted to 1597 cm<sup>-1</sup> at t<br>photoresist coating, (ii) Photolithography, (iii) electrochemical blue-shifted to 2692 cm<sup>-1</sup>,<br>brominating. doping<sup>[24]</sup>, photoresist coating, (ii) Photolithography, (iii) brominating.

 coated on the SLG wafer to increase the adhesion between SLG and photoresist. Second, a thin layer of photoresist (AZ5214E) film was spin-coated on the substrate. Then, the photoresist film was exposed by the laser-beam direct- writing photolithography. Later on, the exposed photoresist was removed by tetramethylammonium hydroxide for several times and the aimed SLG micropattern array was obtained. After that, the obtained SLG micropattern array 9 was immersed in an aqueous electrolyte containing 10 mM KBr and 10 mM H2SO<sup>4</sup> to performed bromination addition reaction by cyclic voltammetry in a potential range from - 12 0.75 to 1.2 V vs.  $Hg/Hg_2SO_4$  reference electrode with a 13 scan rate of 100 mV/s for certain cycles (See Figure S1). 14 Finally, the brominated SLG (SLGBr) micropatterns were 15 obtained after removing the photoresist protection layer by rinsing with acetone and isopropanol.

17 As reported very recently by us, the functionalization 18 of SLG is realized by the bromination addition reaction<sup>[20]</sup> 19 Because SLG is conductive enough to act as the working 20 electrode, bromide anions are oxidized to bromine radicals. 21 which can react with SLG to form SLGBr. When SLG 22 changes into SLGBr, the interfacial electrochemical reaction 23 activity will be decreased because the Faraday current 24 keeping decreasing with the potential scanning going on.25 The SLGBr micropatterns "XMU", the acronym of Xiamen 26 University, were observed to have a different colour with 27 SLG by optical microscopy (Figure 1a). The corresponding 28 images of scanning electron microscopy (SEM) and 29 scanning Raman microscopy (SRM) were shown in Figure 30 1b and 1c. The clear boundaries between the SLG and 31 SLGBr were also observed, indicating the different 32 properties between them.

The Raman spectrum of the SLGBr region is much different from that of the SLG region (Figure 1d). The pristine SLG has a G-peak at near 1586 cm<sup>-1</sup> and a 2D peak at 2687 cm<sup>-1</sup>. The 2Dpeak is sharp and symmetrical, and the peak intensity is about 4 times larger than that of G-peak, indicating that the pristine SLG has a perfect lattice structure<sup>[22]</sup>. After the functionalization by bromination addition reaction, the newly observed D-peak (~1350 cm<sup>-1</sup>) and D'-peak (~1620 cm<sup>-1</sup>) represents the sp<sup>3</sup>-C defects<sup>[23]</sup>, indicating that the symmetry of the lattice is changed 42 due to the formation of C-Br bond. Furthermore, the G-peak<br>Scheme 1 Preparing processes of brominated SLG with patterns. (i) blue objeted to 1507 cm<sup>-1</sup> at the brominated region, and 2D pook with patterns. (i) blue-shifted to 1597 cm<sup>-1</sup> at the brominated region, and 2D peak<br>electrochemical  $\mu_{\text{tot}}$  at the 2000 peak in the theory is the state of  $\Omega_{\text{tot}}$ blue-shifted to 2692 cm<sup>-1</sup>, predicting that SLGBr is p-type doping<sup>[24]</sup>, which is consistent with the strong electron absorption





**Figure 1** The optical image (a) and SEM image (b) of patterned SLGBr with the "XMU" logo.(c) Raman  $I_D/I_G$  mapping of the patterned SLGBr. (d) The corresponding Raman spectra of pristine SLG and SLGBr. (e) Statistical *I<sub>D</sub>/I<sub>G</sub>* histograms from the mapping image of (c).



 The intensity ratio (*I*D/*I*G) is usually adopted to characterize the defect density of graphene, that is, the 3 ratio of the sp<sup>3</sup> to sp<sup>2</sup> bonding character. A higher the  $I_D/I_G$  doping due to the 4 ratio means a higher defect density<sup>[25]</sup>. The statistical the SL 5 analysis of SRM image gives a distribution of *I<sub>D</sub>*/*I<sub>G</sub>* ratio as shown in Figure 1c. The *I*D/*I*<sup>G</sup> ratio in the "XMU" region is much higher than that in the remaining region, which indicates that photoresist micropattern can protect well the underlying graphene and let the bromination addition reaction selectively occur in the exposed area. The statistical *I*D/*I*G-ratio histogram was employed to analyzed 12 the bromination degree of SLG. From Figure 1e,  $I_D/I_G$  ranged from 0.5 to 1.5 with a mid-value about 1.2, suggesting that the bromination degree of SLGBr micropattern was pretty uniform.

 To demonstrate the versatility and reliability of the method, SLGBr micropattern array such as square ring, cross and an H- type Chinese knot were fabricated. The optical microscopy images of the SLG micropatterns after laser-beam direct-writing photolithography were shown in Figure 2a-2c, and the relevant optical images of the SLGBr 22 micropatterns were shown in Figure. S2, indicating the clear 23 SLG/SLGBr boundaries. The relevant *I<sub>D</sub>/I<sub>G</sub>* distribution of 24 the SLGBr micropatterned SLG was characterized by the SRM as shown in Figure 2d-2f, indicating the uniform functionalization of bromination addition reaction.

27 We have demonstrated that the electron transport (i.e., 28 electronic properties) and interfacial electron transfer (i.e., 29 electrochemical properties) of functionalized SLG will change a  $30$  lot from the pristine SLG<sup>[20]</sup>. To investigate the electron transport 31 properties of SLGBr, we designed and fabricated the SLG-32 SLGBr interdigitated electrodes with a pair of Cr/Au belts as the 33 current collectors (Scheme 2) to construct a field-effect transistor 34 (FET), where the SLG electrode acts as the drain, the SLGBr **Communist Communist Communis** 35 acts as the source, and the Si/SiO<sub>2</sub> substrate acts as the gate. In **the state of a state of a state**  $\frac{1}{2}$ 36 order to investigate the characteristics of the devices, SLG **order of a gramma** 



Figure 3 (a-c) The  $I_{DS}$ - $V_{DS}$  curves of the SLGBr Interdigitated Electrode<br>when  $I_D/I_G$  value is 0.5, 1.7 and 2.4, respectively. Inset: the slope of the<br> $I_{DS}$ - $V_{DS}$  curves changes with  $V_{GS}$ . (d) The statistic relati  $I_{DS}$ - $V_{DS}$  curves changes with  $V_{GS}$ . (d) The statistic relationship curve at  $V_{GS}=0$  V.

37 electrodes and SLGBr electrodes were prepared simultaneously. The SLG electrodes exhibit resistance characteristics and p doping due to the residual photoresist (Figure S 3a-3b), while the SLGBr electrodes with higher defect density showed semiconducting characteristics (Figure S 3c-3d). Akin to electrochemically induced In bromination addition introduced new  $sp^3$  C-Br bond to SLG. Because of the asymmetric broken honeycomb structure, the bandgap of SLGBr rises, and a semimetal–semiconductor transition takes place in SLGBr.



Figure 2 (a-c) Optical images of SLG with photoresist patterns, including (a square rings, (b) crosses and (c) H- type Chinese knot. (d-f) Corresponding



**Scheme 2** (a-g) Preparing processes of the graphene Interdigitated Electrodes with designed SLGBr patterns. (h) The optical image of a typical Interdigitated Electrode with patterned SLGBr.

1 Figure 3a-3c show the  $I_{DS}$ - $V_{DS}$  curves of the SLGBr/SLG FET 2 with different bromination degree characterized by the statistical 3 *I*D/*I*<sup>G</sup> mid-values from the SRM images of SLGBr. With the low 4 bromination degree  $(I_D/I_G: 0.5$  in Figure 3a), the  $I_{DS}$ - $V_{DS}$  curves 5 shows a typical electronic property of a resistor. The 6 conductance, i.e., the slope of the *I*<sub>DS</sub>-*V*<sub>DS</sub> curves changes 7 linearly with  $V_{GS}$  in a potential region  $[-50 \text{ V}, +50 \text{ V}]$ , shown as 8 the insert of Figure 3a. The voltage-sensitive feature 9 demonstrates the potential application of SLG into micro-varistor. 10 With the higher bromination degree ( $I_D/I_G$ : 1.7 in Figure 3b or 2.4 11 in Figure 3c), the *I*<sub>DS</sub>-V<sub>DS</sub> curves were obviously different from 12 those in Figure 3a, presenting the characteristics of current 13 rectification. The results show that the SLGBr will be changed 14 into a semiconductor if the bromination degree becomes 15 higher<sup>[20, 26]</sup>. This result is similar to our previous study, but due **Example 1916**. First Rev. 16 to the low degree of bromination, the semiconductor 17 characteristics are less significant than before<sup>[23]</sup>. We performed 18 the electronic measurements of the FETs with different 19 bromination degree and different V<sub>GS</sub> bias, and the statistic 20 relationship between the conductance  $(d_{DS}/dV_{DS})$  and the 21 bromination degree  $(I_D/I_G)$  is shown in Figure 3d. The critical  $I_D/I_G$ 22 value between conductor and semiconductor can be estimated 23 as 0.6. The mechanisms of the change of electron transport 24 properties needs to be further studied in future.<br>25 **big no conclusion, laser beam direct-writing photo-**

In conclusion, laser beam direct-writing photolithography and 26 electrochemically induced bromination addition reaction were 27 adopted for the micropatterning and functionalization of SLG. 28 We found that the brominating functionalization can tune the 29 electron transport properties of SLG by controlling its 30 bromination degree. The conductance of SLGBr keeps 31 decreasing at low bromination degree. After a critical value, 32 SLGBr will become a semiconductor. This work demonstrates 33 the possibility for the fabrication of ultrathin microelectronic 34 devices by SLG.

### 35 **Supporting Information**

36 Supporting information for this article is given via a link at the 37 end of the document.((Please delete this text if not appropriate.))

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## **Entry for the Table of Contents**



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# <sup>2</sup> 单层石墨烯微米尺度图案化和功能化:调控电子传  $\hat{m}$  特性 4 s 崔苗苗 a,韩联欢 a,b\*,曾兰平 a,郭佳瑶 a,宋维英 a,刘川 a,吴元菲 a,罗 6 世翊 <sup>a</sup>, 刘云华 <sup>c\*</sup>, 詹东平 <sup>a\*</sup> 7 (a. 化学化工学院化学系;固体表面物理化学国家重点实验室, 电化学技术教育部工程研究中心, 厦门大学, 8 厦门 361005 9 b 厦门大学萨本栋微纳米科学与技术研究所机电工程系, 厦门大学, 厦门 361005 10 c 华中科技大学国家 CAD 支撑软件工程研究中心, 武汉 430074) 11 12 滴要:石墨烯具有优异的物理特性,如单原子厚度、极高的载流子迁移率等。然而,其零带隙的半金属特性限制了 13 题在高性能场效应晶体管中的应用。为此,研究者们提出了石墨烯纳米化、外场诱导、掺杂以及化学图案化等策略, 14 以调控其带隙宽度。但是,这些方法的可控性以及稳定性还需要进一步改善。在本研究中,我们提出采用电化学溴 15 化并结合光刻图案化调控单层石墨烯的电子传输特性。通过这种方法,我们成功制备了图案化的溴化石墨烯 16 (SLGBr)。进一步研究表明单层石墨烯的电子传输性能可以通过溴化程度来调控。当溴化程度较小时,SLGBr 表 17 现为电阻特性, 且其电导随溴化程度增加而减小; 当溴化程度增加到一定值时, SLGBr 表现为与半导体类似的特性。 18 本研究将为全石墨烯器件的制备提供可行的技术路线,拓展其在微电子领域的应用。

19

20 关键词: 石墨烯图案化; 电子传输; 电化学溴化; 光刻; 全石墨烯器件